## DESIGN AND DEVELOPMENT OF FERRITE-TIO<sub>2</sub> NANOCOMPOSITES WITH TUNABLE MAGNETIC PROPERTIES

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Ni-ferrite-TiO<sub>2</sub> nanocomposites with varying TiO<sub>2</sub> content (0%, 25%, 50% and 75%) were synthesized using the sol-gel autocombustion method and characterized through XRD, FE-SEM, VSM, and Raman spectroscopy. The XRD analysis confirmed the coexistence of ferrite and TiO<sub>2</sub> phases. FE-SEM images revealed uniform particle distribution and a reduction in particle size as TiO<sub>2</sub> content increased. Raman spectroscopy showed strong TiO<sub>2</sub>-related vibrational modes, with the highest intensity observed in the 75% TiO<sub>2</sub> sample, diminishing as TiO<sub>2</sub> content decreased. Peaks observed in pure Ni-ferrite (283, 402, 469 and 689 cm<sup>-1</sup>) shifted to lower wavelengths with increasing TiO<sub>2</sub> doping, indicating altered vibrational modes due to phase interactions. These interactions likely contributed to changes in the magnetic properties. VSM analysis revealed a decrease in saturation magnetization and magnetic remanence with increasing TiO<sub>2</sub> content, while coercivity remained stable. The magnetic behavior was attributed to TiO<sub>2</sub> dilution and phase interfaces, offering valuable insights for the design of magnetic materials with customized properties.

**Keywords:** Ferrite; Nanocomposites; Magnetic properties; Nanoparticles; Hybrid ferrite

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#### 1. INTRODUCTION

The pursuit of advanced magnetic materials with tailored properties has become a central focus of modern technological innovation. Magnetic nanomaterials offer promising opportunities for applications in spintronics, magnetic storage, biomedical devices, and energy storage systems [1][2]. Among these, nickel ferrite (NiFe) nanoparticles stand out due to their remarkable magnetic characteristics, stability, and biocompatibility [3].

Ni ferrite nanoparticles are known for their excellent magnetic properties, such as high saturation magnetization and coercivity, making them suitable for various high-performance applications [4,5]. Moreover, the magnetic behavior of these nanoparticles can be further optimized by altering their composition, structure, and morphology [6]. Studies on similar ferrite-based nanocomposites, such as cobalt ferrite (CoFe) and zinc ferrite (ZnFe), have demonstrated distinct magnetic properties. CoFe nanoparticles exhibit high magnetic anisotropy and coercivity [7], while ZnFe nanoparticles are recognized for their magnetic stability and biocompatibility [8].

The potential of NiFe nanoparticles can be expanded through hybridization with other materials, creating multifunctional nanocomposites with enhanced magnetic and structural properties. Titanium dioxide ( $TiO_2$ ) emerges as an excellent material for this purpose due to its unique optical, electrical, and magnetic properties [9]. Integrating  $TiO_2$  with NiFe nanoparticles can yield nanocomposites with enhanced magnetic properties, improved thermal stability, and increased surface area [10].

In this study, Ni-ferrite-TiO<sub>2</sub> nanocomposites with varying TiO<sub>2</sub> content were synthesized using the sol-gel autocombustion method. This technique provides precise control over the composition, structure, and morphology of the nanocomposites, enabling a systematic investigation of how TiO<sub>2</sub> incorporation influences their magnetic properties [11]. Our research investigates the structural, morphological, and magnetic properties of NiFe-TiO<sub>2</sub> nanocomposites to gain a deeper understanding of the mechanisms underlying their enhanced magnetic behavior. The insights gained from this study will aid in the development of multifunctional nanomaterials with customized magnetic properties, opening new possibilities for applications in magnetic storage, spintronics, biomedical devices, and energy storage technologies.

### 2. Materials and methods 2.1 Synthesis

NiFe<sub>2</sub>O<sub>4</sub> spinel nanoferrite was synthesized using a sol-gel auto-combustion method [12,13]. Stoichiometric amounts of analytical grade of ferric nitrate (Fe(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O), nickel nitrate (Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O), and citric acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>) were dissolved in deionized water, maintaining a 1:1 molar ratio [14]. The pH was adjusted to neutral (pH 7) by adding ammonia while stirring. The mixture was then heated to 90°C, forming a dense gel, which was subsequently heated to 275°C, initiating a combustion reaction that produced as-burnt ferrite powder. The powder was calcined at 500°C for 3

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hours in air to remove organic residues and achieve a uniform particle size distribution [15]. To explore the impact of TiO<sub>2</sub> content on the nanocomposite properties, TiO<sub>2</sub> nanoparticles were introduced into the ferrite nanopowder at various ratios (0%, 25%, 50%, and 75%). The structural, morphological, and magnetic properties of the resulting ferrite-TiO<sub>2</sub> nanocomposites were comprehensively characterized.

#### 2.2 Material characterization

The crystalline structure of the ferrite nanocomposites was examined using X-ray diffraction (XRD) analysis performed on a PANalytical X'pert Pro instrument (Netherlands) with a Cu K $\alpha$  radiation source ( $\lambda$  = 1.54048 Å) [16]. The surface morphology of the nanoparticles was investigated using field emission-scanning electron microscopy (FE-SEM) with a MIRA3-XMU model (TESCAN, Czech Republic) [17]. Raman spectra using XploRa-HOBIBA with a 785 nm laser were employed to identify the spinel ferrite phase [18]. The magnetic properties of the ferrite nanocomposites were characterized at room temperature using a vibrating sample magnetometer (VSM) from LBKFB Meghnatis Daghigh Kavir Company, with an applied magnetic field ranging from -10 to +10 kOe [19].

#### 3. RESULTS AND DISCUSSION

The XRD patterns of the NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites with varying TiO<sub>2</sub> content (0%, 25%, 50%, and 75%) are presented in Figure 1. The FullProf-fitted spectra reveal well-defined peaks at (220), (311), (222), (400), (422), and (440), which are characteristic of the spinel ferrite phase [20]. These peaks confirm the presence of the NiFe<sub>2</sub>O<sub>4</sub> spinel structure and indicate that the samples are free from detectable impurities [21]. In addition to the ferrite phase, the TiO<sub>2</sub> phase was identified through spectral fitting techniques [22]. The XRD patterns correspond well with known reference patterns for the cubic spinel phase of NiFe<sub>2</sub>O<sub>4</sub> (JCPDS card no. 00-054-0964) and the tetragonal anatase phase of TiO<sub>2</sub> (JCPDS card no. 00-021-1272) [23]. The accurate matching of these patterns with standard reference cards validates the successful incorporation of TiO<sub>2</sub> into the NiFe<sub>2</sub>O<sub>4</sub> matrix and confirms the preservation of the distinct crystal structures of both components within the nanocomposites. This indicates that TiO<sub>2</sub> is present in its anatase form and does not interfere with the spinel structure of the ferrite, thus allowing for the exploration of combined properties of these materials.

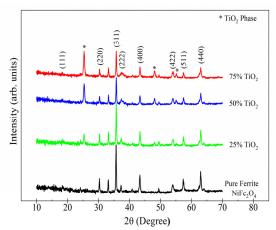


Figure 1. XRD patterns of 0%, 25%, 50%, and 75% NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites

The structural parameters of the NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nano composites, determined through FullProf fitting, are summarized in Table 1. The crystallite size ( $D_{XRD}$ ) and particle size ( $D_{FE-SEM}$ ) values show a slight variation with increasing TiO<sub>2</sub> content. Specifically, a minimum crystallite size of 41.72 nm is observed for the 25% TiO<sub>2</sub> sample. This reduction in crystallite size can be attributed to the grain refinement effect induced by the incorporation of TiO<sub>2</sub>. The presence of TiO<sub>2</sub> may disrupt the growth of NiFe<sub>2</sub>O<sub>4</sub> grains, leading to smaller crystallite sizes [24]. The lattice parameter (a) values range from 5.347 Å to 7.213 Å, reflecting a subtle expansion of the crystal lattice with increasing TiO<sub>2</sub> content. This lattice expansion is likely due to the substitution of larger Ti<sup>4+</sup> ions for the smaller Fe<sup>3+</sup> ions in the NiFe<sub>2</sub>O<sub>4</sub> lattice. As Ti<sup>4+</sup> ions are larger than Fe<sup>3+</sup> ions, their incorporation into the NiFe<sub>2</sub>O<sub>4</sub> lattice increases the lattice dimensions, which is consistent with the observed increase in the lattice parameter [25].

**Table 1.** Structural Parameters from XRD & SEM analysis- crystallite size  $(D_X)$ , particle size  $(D_S)$ , lattice parameter (a), density  $(\rho x)$ , microstrain( $\epsilon$ ) and dislocation density  $(\delta)$ 

TiO <sub>2</sub> %	D <sub>XRD</sub> (nm)	D <sub>SEM</sub> (nm)	<i>a</i> (Å)	$\rho_{x}$	ε×10 <sup>-3</sup>	δ×10 <sup>15</sup>
0	44.85	17.9	8.350	5.35	1.5	0.599
25	41.72	19.9	8.335	7.21	1.91	0.824
50	45.01	22.1	8.333	7.21	1.80	0.720
75	49.08	21.9	8.335	7.21	1.61	0.591

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The density  $(\rho_x)$  of the nanocomposites is found to be lowest for the 0% TiO<sub>2</sub> sample and remains relatively constant for the composites with higher TiO<sub>2</sub> content. This suggests that the addition of TiO<sub>2</sub> does not significantly alter the overall density of the nanocomposites. The lack of significant change in density might indicate that the TiO<sub>2</sub> particles are well-dispersed within the NiFe<sub>2</sub>O<sub>4</sub> matrix, without causing substantial changes in the overall packing density of the material [25]. Additionally, the microstrain  $(\epsilon \times 10^{-3})$  values decrease with increasing TiO<sub>2</sub> content, indicating a relaxation of the crystal structure. Microstrain is a measure of local lattice distortions and defects. The decrease in microstrain with higher TiO<sub>2</sub> content suggests that the incorporation of TiO<sub>2</sub> reduces lattice distortions and helps to relieve internal stresses within the NiFe<sub>2</sub>O<sub>4</sub> lattice. This reduction in microstrain is likely due to the reduced lattice mismatch between the NiFe<sub>2</sub>O<sub>4</sub> and TiO<sub>2</sub> phases, leading to a more coherent and less strained crystal structure [26].

The surface morphology of the NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nano composites was examined using field emission scanning electron microscopy (FESEM), with the resulting images presented in Figure 2a-d. The FESEM images provide detailed insights into the particle size, shape, and distribution of the nanocomposites. These images reveal that the nanoparticles exhibit a relatively uniform distribution across the sample, with variations in particle size that are dependent on the TiO<sub>2</sub> content. The observations from the FESEM analysis are crucial for understanding the impact of TiO<sub>2</sub> incorporation on the surface characteristics of the NiFe<sub>2</sub>O<sub>4</sub> matrix, which can influence the material's overall performance in various applications [27].

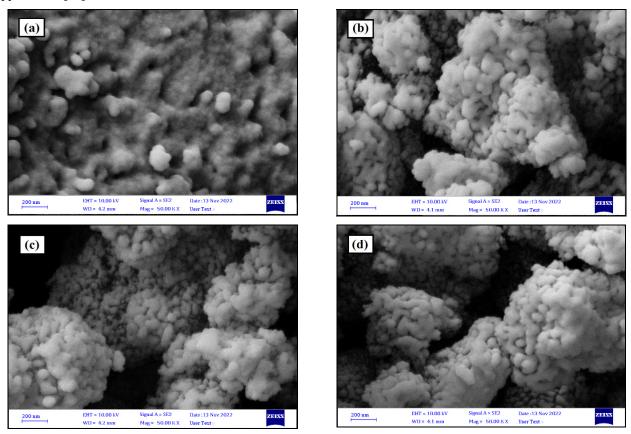


Figure 2. FE-SEM images of 0%, 25%, 50%, and 75% NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites

The FE-SEM microstructural analysis (Figure 3a-d) revealed a nearly homogeneous particle size distribution across the surface, with particles predominantly exhibiting semi-spherical shapes. This analysis highlights the significant effect of TiO<sub>2</sub> doping on particle size, demonstrating that higher TiO<sub>2</sub> content (75%) results in noticeable particle enlargement. The microstructural examination provides detailed insights into the average size and grain growth characteristics, which are essential for understanding the material's physical, electrical, and magnetic properties. Particle agglomeration is evident, with smaller particles coalescing to achieve a lower free energy state, a trend that is more pronounced with increased TiO<sub>2</sub> doping. Additionally, there is a noticeable tendency for aggregation among numerous ferrite nanoparticles. The observed porous structures may result from substantial gas release during the combustion process, while the formation of agglomerated regions is likely due to the inherent interactions among magnetic nanoparticles [28].

The average crystallite size ( $D_{XRD}$ ) and particle size ( $D_{FE-SEM}$ ) show a significant dependence on TiO<sub>2</sub> content, as summarized in Table 1. The images also indicate that the particle sizes of the two phases vary with the relative content of each component. As TiO<sub>2</sub> content increases, the size of the nanocomposite particles also increases. Uniform structures are observed in the intermediate compositions, suggesting coherent grain growth within the constituent phases of the composites.

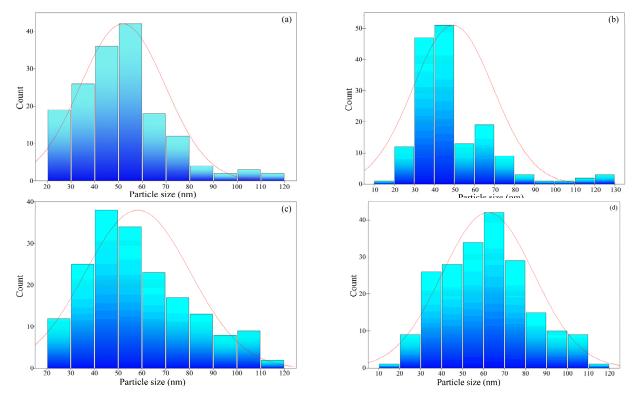


Figure 3. FE-SEM images of 0%, 25%, 50%, and 75% NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites

Figure 4 presents the Raman spectra of NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites with varying TiO<sub>2</sub> content (0%, 25%, 50%, and 75%). The Raman spectroscopy results reveal a prominent vibrational mode associated with the TiO<sub>2</sub> phase, as evidenced by the high-intensity peaks observed in the 75% TiO<sub>2</sub> sample [29]. As the TiO<sub>2</sub> content decreases, the intensity of these TiO<sub>2</sub>-related peaks diminishes, highlighting the influence of TiO<sub>2</sub> concentration on the Raman signal. In contrast, the pure Ni-ferrite sample exhibits characteristic peaks at 283, 402, 469, and 689 cm<sup>-1</sup>. These peaks shift towards lower wavelengths with increasing TiO<sub>2</sub> doping percentages, suggesting changes in the vibrational modes of the Ni-ferrite phase due to interactions with the TiO<sub>2</sub>. The observed shift in peak positions and variations in peak intensities indicate a strong interaction between the Ni-ferrite and TiO<sub>2</sub> phases. This interaction likely influences the material's vibrational properties and is correlated with changes in the magnetic properties of the nanocomposites as TiO<sub>2</sub> content varies [30].

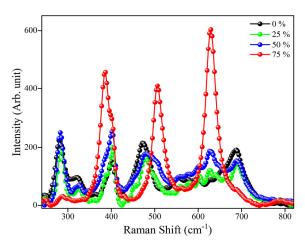


Figure 4. Raman spectra of 0%, 25%, 50%, and 75% NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites

Figure 5 displays the room temperature hysteresis loops derived from Vibrating Sample Magnetometry (VSM) measurements. The magnetic properties, saturation magnetization (Ms), coercivity (Hc), and remanence magnetization ( $M_R$ ), are summarized in Table 2.

Figure 6 presents the variation curves for Ms and  $M_R$  with different TiO<sub>2</sub> concentrations. The results reveal a decline in Ms and magnetic moment ( $\mu$ ) as TiO<sub>2</sub> content increases, indicating a systematic reduction in the magnetic properties of the NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites. This reduction is attributed to the dilution effect caused by the incorporation of TiO<sub>2</sub>, which progressively diminishes the magnetic properties of the NiFe<sub>2</sub>O<sub>4</sub> phase [31,32]. Since TiO<sub>2</sub>

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is non-magnetic, its presence reduces the concentration of the magnetic NiFe<sub>2</sub>O<sub>4</sub> phase, leading to an overall decrease in the magnetic moment of the nanocomposites. Notably, Hc remains relatively stable across different TiO<sub>2</sub> concentrations, suggesting that the incorporation of TiO<sub>2</sub> does not significantly affect the magnetic anisotropy of the nanocomposites [33,34]. The consistent coercivity indicates that TiO<sub>2</sub> does not substantially alter the magnetic domain structure or the magnetization reversal process in the NiFe<sub>2</sub>O<sub>4</sub> phase.

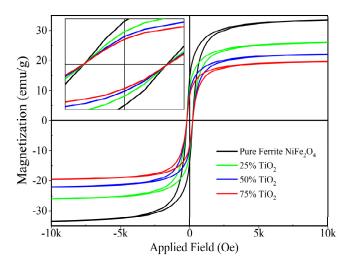


Figure 5. RT Magnetic curves of 0%, 25%, 50%, and 75% NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites

Table 2. RT Magnetic parameters

TiO <sub>2</sub>	Ms	$M_R$	Hc	ηB	K×10 <sup>3</sup>
%	(emu/g)	(emu/g)	(Oe)	(µB)	(emu.Oe/g)
0	33.73	14.29	204.1	1.416	7.17
25	26.28	10.86	204.7	1.479	5.60
50	22.24	9.19	205.9	1.251	4.77
75	19.75	8.3	206.9	1.111	4.26

The observed decrease in saturation magnetization and magnetic moment with increasing TiO<sub>2</sub> content may also be related to changes in particle size and surface effects, as indicated by the FESEM analysis [35]. Variations in particle size and potential surface phenomena, such as increased porosity or agglomeration, could further impact the magnetic properties of the nanocomposites. These factors must be considered when interpreting the changes in magnetic behavior with varying TiO<sub>2</sub> concentrations.

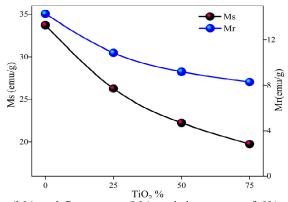


Figure 6. RT Magnetic Saturation (Ms) and Remanence ( $M_R$ ) variation curves of 0%, 25%, 50%, and 75% NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites

### 4. CONCLUSIONS

This study provides significant insights into NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites, focusing particularly on their magnetic properties alongside structural and morphological aspects. X-ray diffraction (XRD) confirmed the successful synthesis of these nanocomposites, revealing the presence of both NiFe<sub>2</sub>O<sub>4</sub> and TiO<sub>2</sub> phases. Field emission scanning electron microscopy (FESEM) demonstrated a uniform distribution of TiO<sub>2</sub> within the NiFe<sub>2</sub>O<sub>4</sub> matrix, with particle size decreasing as TiO<sub>2</sub> content increased, suggesting that TiO<sub>2</sub> acts as a grain refiner. Raman spectroscopy further revealed strong interactions between NiFe<sub>2</sub>O<sub>4</sub> and TiO<sub>2</sub>, leading to changes in vibrational modes that influence the material's magnetic behavior.

The magnetic studies, conducted via Vibrating Sample Magnetometry (VSM), are particularly noteworthy. The results showed a significant decrease in saturation magnetization and magnetic moment as TiO<sub>2</sub> content increased, highlighting a pronounced dilution effect on the magnetic properties of the NiFe<sub>2</sub>O<sub>4</sub> phase. This reduction is a direct consequence of the non-magnetic TiO<sub>2</sub> phase diluting the magnetic contribution of NiFe<sub>2</sub>O<sub>4</sub>. Despite this decrease, the coercivity remained stable across different TiO<sub>2</sub> concentrations, indicating that TiO<sub>2</sub> incorporation does not significantly affect the magnetic anisotropy or the magnetization reversal process. This stability in coercivity suggests that the fundamental magnetic characteristics of the NiFe<sub>2</sub>O<sub>4</sub> phase are preserved, even as its magnetic moment is diluted.

Looking ahead, future research should prioritize optimizing synthesis parameters to further refine these magnetic properties, exploring different TiO<sub>2</sub> doping levels and alternative materials. Additionally, practical applications in spintronics, magnetic sensors, and catalysis could benefit from these findings, making it crucial to evaluate how these nanocomposites perform in specific technological contexts. Comparative studies with other materials will also be valuable to highlight the relative advantages of NiFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanocomposites. Overall, this work advances our understanding of TiO<sub>2</sub> doping effects on NiFe<sub>2</sub>O<sub>4</sub> nanocomposites, emphasizing their potential for applications requiring tailored magnetic properties.

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#### **Conflict of interest**

On behalf of all authors, the corresponding author states that there is no conflict of interest.

#### Data access statement

All data generated during this study are included in the manuscript.

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### ПРОЕКТУВАННЯ ТА РОЗРОБКА НАНОКОМПОЗИТІВ ФЕРИТ-Ті $O_2$ З НАСТРОЮВАНИМИ МАГНІТНИМИ ВЛАСТИВОСТЯМИ

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Нанокомпозити Ni-ферит-TiO<sub>2</sub> з різним вмістом TiO<sub>2</sub> (0%, 25%, 50% та 75%) були синтезовані за допомогою методу зольгель автогоріння та охарактеризовані за допомогою рентгенівської дифракції (XRD), FE-SEM, VSM та раманівської спектроскопії. Рентгенівський дифракціональний аналіз підтвердив співіснування фаз фериту та TiO<sub>2</sub>. Зображення FE-SEM показали рівномірний розподіл частинок та зменшення розміру частинок зі збільшенням вмісту TiO<sub>2</sub>. Раманівська спектроскопія продемонструвала сильні коливальні моди, пов'язані з TiO<sub>2</sub>, причому найвища інтенсивність спостерігалася у зразку з 75% TiO<sub>2</sub>, зменшуючи її зі зменшенням вмісту TiO<sub>2</sub>. Піки, що спостерігалися в чистому Ni-фериті (283, 402, 469 та 689 см<sup>-1</sup>), зміщувалися до нижчих довжин хвиль зі збільшенням легування TiO<sub>2</sub>, що вказує на змінені коливальні моди через фазові взаємодії. Ці взаємодії, ймовірно, сприяли змінам магнітних властивостей. Аналіз VSM виявив зменшення намагніченості насичення та залишкової магнітної напруги зі збільшенням вмісту TiO<sub>2</sub>, тоді як коерцитивна сила залишалася стабільною. Магнітна поведінка була пов'язана з розведенням TiO<sub>2</sub> та межами розділу фаз, що дає цінні знання для розробки магнітних матеріалів з індивідуальними властивостями.

Ключові слова: ферит; нанокомпозити; магнітні властивості; наночастинки; гібридний ферит